### Comparison of the Electrochemical and Physical Properties of Nanocrystalline Copper Deposition in the Fabrication of Printed Wiring Boards

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#### Abstract

Typical electrodeposition of conventional metals produces deposits that are polycrystalline in nature, comprised of many crystal grains separated by grain boundaries. Adding grain refiners to a plating solution and employing pulseplating techniques can reduce the grain size and produce a nanocrystalline deposit. The average grain size of the nanocrystalline copper deposit is about 100 nanometers. This is about 80 times smaller than the conventional deposit average grain size of 2 microns. Nanocrystalline copper deposits have negligible porosity and superior physical, mechanical, and electrical properties. The hardness, strength and wear resistance of the deposit are greatly enhanced. Stress corrosion cracking is virtually eliminated, while the hydrogen diffusivity and solubility are increased. This paper compares the electrochemical, mechanical, and physical properties of nanocrystalline copper deposits with conventional polycrystalline copper deposits on printed wiring boards (PWB). Test boards were evaluated after thermal shock and thermal stress tests. Copper thickness and uniformity are evaluated both by microsection and X-ray fluorescence measurement techniques.

#### Introduction

The electrodeposition of nanocrystalline metals has attracted considerable interest due to their improved electrochemical, mechanical and physical properties. It has been demonstrated that nanocrystalline deposits produced by pulsed electro deposition (PED) have a higher hardness, lower friction coefficient and lower electrical resistance compared to polycrystalline deposits produced by direct current (DC) plating [1].

The deposition of nanostructure deposits by PED is possible by optimizing the pulse length (time on), the time between two pulses (time off), the peak height (pulse), and the average current density [2]. Pulse electro-deposition permits electrolysis with a high current density during a short period of time [3].

The addition of organic additives such as complex formers and inhibitors are also necessary to achieve smaller grains. These additives aid in inhibiting crystallite growth resulting in a finer grained structure.

This paper compares the mechanical and physical properties of PED nanocrystalline copper deposits with DC conventional polycrystalline copper deposits on printed wiring board plated through holes.

#### Background

As advanced printed wiring board (PWB) designs become more complex, the thickness of the board has increased due to the greater number of layers. The plated through hole (PTH) diameters have become much smaller to accommodate the greater density of advanced designs. The increased board thickness results in the PTH's becoming less reliable due to a coefficient of thermal expansion mismatch between the PCB dielectric material and the plated copper barrel during thermal cycling stress.

Printed wiring boards with PTH's are subjected to thermal shock and thermal stress tests to determine the capability of the PTH's to withstand temperature variations.

The objective of this study is to determine if nanocrystalline copper deposits are superior to polycrystalline copper deposits by employing thermal cycle and electrical resistance tests.

High aspect ratio through hole plating tests were also performed to compare the throwing power of each plating process.

#### **Experimental Detail**

Nanocrystalline and polycrystalline copper deposits were produced by PED and DC deposition processes in a copper sulfate electrolyte with citric acid as an additive. Three different bath compositions were used listed in Table 1.

Bath A is a standard copper plating bath consisting of copper sulfate, sulfuric acid and a commercial brightener. Bath B consists of copper sulfate, sulfuric acid and citric acid. Bath C consists of copper sulfate, ammonium sulfate and citric acid. The purpose of Bath A was to produce the industry-typical polycrystalline copper deposits using DC rectification with a current density of 0.138A/in<sup>2</sup>. Baths B and C were used to produce the nanocrystalline copper deposits using PED rectification with a current density of 0.165A/in<sup>2</sup> with a peak current of 10A.

Table 1-Dath Compositions					
Bath	Component	g/l	Additive	g/l	
А	CuSO <sub>4</sub> ·4H <sub>2</sub> O H <sub>2</sub> SO <sub>4</sub>	80 225	brightener	25	
	CuSO <sub>4</sub> ·4H <sub>2</sub> O	28	citric acid	30	
В	$H_2SO_4$	225	brightener	25	
C	CuSO <sub>4</sub> ·4H <sub>2</sub> O	28		50	
U	(NH4) <sub>2</sub> SO <sub>4</sub>	50	chric acid	50	

This study used test coupons designed specifically for aspect ratio plating test and thermal cycling test. The test coupons were 62 and 93 mil thick polyimide boards. The plated copper thickness in the through holes was 1.25-1.5 mils. 12" X 18" panels were drilled and electroless copper was deposited. The panels were then routed to 2"x 4" test coupons. The test coupons were panel plated then patterned with dry film photoresist and etched.

All experiments were conducted using a 5 gallon plating tank. Agitation was supplied by a recirculation filter pump with a center tank sparger. Oxygen Free High Conductivity copper anodes were and anode bags were used. A Kraft Dynatronix model DP20-5-10 power supply was used.

Analysis of the deposits included SEM micrographs, hardness tests, electrical resistance tests, thermal shock tests, and thermal stress tests.

### **Deposit Analysis**

Scanning Electron Microscopy (SEM) was used to determine the deposit morphology. The deposits from bath A have typical polycrystalline grain structure with an average grain size of about 2 microns (Figure 1 and 2).



Figure 1 – Bath A



Figure 2 – Bath A

The deposits from bath B have a nanocrystalline grain structure with an average grain size of about 150 nm (Figure 3 and 4).



Figure 4 – Bath B

The deposits from bath C have a nanocrystalline grain structure with an average grain size of about 50 nm (Figure 5 and 6).



Figure 5 – Bath C



Figure 6 – Bath C

#### **Hardness Test**

Hardness tests were performed with a Clark MHT-1 microhardness tester using a Vickers diamond pyramid indenter with an applied force of 50g for 10 seconds. A low force was selected to isolate measurement to the plating layer only. The hardness measurements in Table 2 show as the grain size is reduced the deposit becomes harder.

Bath	Grain size(nm)	Force (g)	Hardness (GPa)
Bath A	2000	50	0.49
Bath B	150	50	1.21
Bath C	50	50	1.56

### 

#### **Thermal Shock Test**

Thermal shock tests per MIL-PRF-55110G and IPC-TM-650 were performed on test coupons from each bath. The thermal cycle test board consists of 140 PTH's connected through a daisy chained pattern (Figure-7). The hole size is 28 mils with a 55 mil pad in a 62 mil thick polyimide board. The glass transition temperature of polyimide is 260°C.



Figure 7 – Thermal Shock Test Coupon

The test specimens were subjected to 90 temperature cycles listed in table 3. The requirements of IPC-TM-650 state a high temperature limit of 170°C for polyimide dielectric material. For this test the high temperature was elevated to 177°C (the temperature limit of the test chamber) to increase the thermal expansion stress in the PTH's.

Table 5- Thermal Shock Temperature Cycle					
Low	Dwell	Dwell			
Temperature	Time	Temperature	Time		
-65°C	15 (min)	+177°C	15 (min)		

### Table 3. Thermal Shack Temperature Cycle

Interconnection resistance measurements were taken every minute during the thermal shock test. The test results listed in Table 4 indicate that after thermal cycling, test coupons from each bath passed the continuity test and there was no indication of an open circuit. Test coupons from Bath A and Bath B had a total resistances change of more than 10 percent between the first high temperature cycle and the last high temperature cycle. The requirements of IPC-TM-650 states that the total resistances change shall not be more than 10 percent. The test coupon from Bath C had a much lower initial resistance then the other baths and remained relatively stable throughout the test.

I ä	1 able 4- Resistance Measurements (onnis)					
Bath	Initial	First high temp.	Last high temp.	Final	Change %	
Α	0.901	1.213	0.882	0.610	28%	
В	0.720	0.379	0.427	0.304	11.5%	
С	0.165	0.244	0.242	0.161	1%	

 Table 4- Resistance Measurements (ohms)

After thermal shock testing, each coupon was microsectioned and visually inspected. No plating cracks, blistering or delamination was observed in the PTH's (Figures 8-10).



Figure 8 – Bath A



Figure 9 – Bath B



Figure 10 – Bath C

#### **Thermal Stress Test**

Test specimens from each bath were subjected to thermal stress tests consisting of an oven bake out at 121°C for a minimum of six hours and a solder dip at 260°C for 10 sec., 20 sec. and 30 sec. Results listed in Table 5 show Bath

A failed at 30 sec., Bath B failed at 20 sec. and Bath C passed all three solder dips. All failures were delamination of the plated copper from the dielectric in the thru hole (Figure-11-13).

Table 5- Therman Stress Test					
Bath	10 sec.	20 sec.	30 sec.		
А	pass	pass	fail		
В	pass	fail	fail		
C	pass	pass	pass		

**Table 5- Thermal Stress Test** 



Figure 11 Bath A 30 sec.



Figure 12 Bath B 20 sec.



Figure 13 Bath C 30 sec.

### High Aspect Ratio Through Hole Plating Test

The high aspect ratio plating test coupon consists of a 93 mil thick polyimide board with seven hole sizes of 8, 10, 15, 20, 28, 40 and 50 mils, representing aspect ratios from 1.25:1 up to 11.37:1(Figure 14).



Figure 14 - Test Coupon

Test coupons were plated in each bath then microsectioned to obtain the surface to hole thickness ratio (SHTR). The SHTR is determined by dividing the plating thickness on the surface of the hole by the plating thickness in the center of the hole. The plated copper thickness in the thru holes was 1.25-1.5 mils. The results listed in Table 6 shows Bath B had the lowest SHTR. Baths A and B reliably plated hole sizes down to 8 mils (Figures 15-16). Bath C had the highest SHTR and did not have the throwing power to deposit the 1.25 mils of copper without closing the 8 and 10 mil holes (Figures 17-18).

Aspect Ratio	Bath A	Bath B	Bath C
11.37:1	2.73	1.97	3.03/closed
9.1:1	2.63	1.88	2.98/closed
6.06:1	2.31	1.78	2.78/
4.55:1	1.94	1.68	2.3
3.25:1	1.79	1.62	2.11
2.27:1	1.65	1.58	1.92
1.82:1	1.55	1.43	1.82

Table 6- Surface to Hole Thickness Ratio



Figure 15- Bath A 8 mil hole



Figure 16- Bath B 8 mil hole



Figure 17- Bath C 8 mil



Figure 18- Bath C 10 mil hole

#### Conclusion

It was demonstrated that the nanocrystalline deposits produced using PED with grain sizes smaller than 100nm have a lower electrical resistance and a higher hardness then polycrystalline copper deposits produced by DC plating. The nanocrystalline deposits passed the elevated temperature of the thermal shock test and the increased dwell time of the thermal stress test.

A nanocrystalline plating process with a brightener additive exhibited a higher throwing power when plating high aspect ratio holes.

It can be concluded that nanocrystalline plating processes can be employed in the fabrication of printed wiring boards and these processes demonstrate superior mechanical and physical properties then conventional polycrystalline plating processes.

#### Acknowledgements

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## Nanocrystalline Metals

Typical direct current (DC) electrodeposition of conventional metals produce deposits that are polycrystalline in nature.

Polycrystalline deposits have an average grain size of 2 microns

Nanocrystalline deposits produced by pulsed electrodeposition (PED) have an average grain size of 100nm





# Mechanical and Physical and Properties of Nanocrystalline Metals

- Nanocrystalline deposits produced by (PED), Compared to polycrystalline deposits produced by direct current (DC) plating have demonstrated...
- Higher hardness
- Lower friction coefficient
- Lower electrical resistance
- Less porosity





- The deposition of nanostructure deposits by PED is possible by optimizing pulse parameters
- Pulse length (time on)
- Time between two pulses (time off)
- Peak height (pulse)
- Average current density



## **Organic Additives**

- The addition of organic additives such as complex formers and inhibitors are also necessary to archive smaller grains.
- These additives aid in inhibiting crystallite growth resulting in a finer grained structure.



- As advanced printed wiring board (PWB) designs become more complex, the thickness of the boards has increased
- The plated through hole (PTH) diameters have become much smaller to accommodate the greater density of advanced designs.
- The increased board thickness results in the PTH's becoming less reliable due to a coefficient of thermal expansion mismatch between the PCB dielectric material and the plated copper during thermal cycling stress.



# **Study Objectives**

- This paper compared the mechanical and physical properties of PED nanocrystalline copper deposits with DC conventional polycrystalline copper deposits on printed wiring board plated through holes
- The objective is to determine if nanocrystalline copper deposits in PTH's are superior to polycrystalline copper deposits by employing thermal cycle and electrical resistance tests
- High aspect ratio through hole plating tests were performed to compare the throwing power of each plating process



## **Experimental Detail**

- Nanocrystalline and polycrystalline copper deposits were produced by PED and DC deposition processes
- Bath A is a standard copper plating bath consisting of copper sulfate, sulfuric acid and a commercial brightener
- Bath B consists of copper sulfate, sulfuric acid, commercial brightener and citric acid
- Bath C consists of copper sulfate, ammonium sulfate and citric acid

Bath	Component	g/l	Additive	g/l
Α	CuSO <sub>4</sub> ·4H <sub>2</sub> O	80	brightener	25
	H <sub>2</sub> SO <sub>4</sub>	225		
В	CuSO <sub>4</sub> ·4H <sub>2</sub> O	28	citric acid	30
	H <sub>2</sub> SO <sub>4</sub>	225	brightener	25
C	CuSO <sub>4</sub> ·4H <sub>2</sub> O	28	citric acid	50
	(NH4) <sub>2</sub> SO <sub>4</sub>	50		



## **Experimental Detail**

- All experiments were conducted in a 5 gallon tank.
- Agitation was supplied by a recirculation filter pump with a center tank sparger.
- Copper anodes were OFHC and anode bags were used.
- Kraft Dynatronix model DP20-5-10 power supply was used.



# **Test Coupons**

- This study used test coupons designed for thermal cycling test and high aspect ratio plating test
- Test coupons were 62 and 91 mil thick polyimide boards
- The plated copper thickness in the through holes was 1.25-1.5 mils
- The boards were drilled, electroless copper plated as a 12"x18" panel
- The panels were then routed to 2"x 4" test coupons
- The test coupons were panel plated then patterned with dry film photoresist and etched



## **Deposit Analysis**

- SEM micrographs
- Hardness tests
- Electrical resistance test
- Thermal shock tests
- Thermal stress tests
- Microsection





Bath A produced polycrystalline copper deposits using DC rectification with a current density of 0.138A/in<sup>2</sup>. Average grain size is 2 microns





Bath B produced nanocrystalline copper deposits using PED rectification with a current density of 0.165A/in<sup>2</sup> with a peak current of 10A.

Average grain size is 150 nm





Bath C produced nanocrystalline copper deposits using PED rectification with a current density of 0.165A/in<sup>2</sup> with a peak current of 10A.

Average grain size is 50 nm





### Hardness Test

 Hardness tests were performed with a Clark MHT-1 microhardness tester using a Vickers diamond pyramid indenter with an applied force of 50g for 10 seconds.

 A low force was selected to isolate measurement to the plating layer only.

 The hardness measurements show as the grain size is reduced the deposit becomes harder

_	Hardness lest Results						
Bath	Grain size (nm)	Force (g)	Hardness (GPa)				
Bath A	2000	50	0.49				
Bath B	150	50	1.21				
Bath C	50	50	1.56				



# **Thermal Shock Test**

- Thermal shock tests per MIL-PRF-55110G and IPC-TM-650 were performed on test coupons from each bath
- The thermal cycle test board consists of 140 PTH's connected through a daisy chained pattern
- The hole size is 28 mils with a 55 mil pad in 62 mil thick polyimide board.
- The glass transition temperature of polyimide is 260° C





# **Thermal Shock Test**

- The test specimens were subjected to 90 temperature cycles.
- The requirements of IPC-TM-650 state a high temperature limit of 170° C for polyimide dielectric material
- For this test the high temperature was elevated to 177°C to increase the thermal expansion stress in the PTH's
- •The test results indicate that after thermal cycling, test coupons from each bath passed the continuity test and there was no indication of an open circuit
- •Test coupons from Bath A and Bath B had a total resistances change of more than 10 percent between the first high temperature cycle and the last high temperature cycle

•The test coupon from Bath C had a much lower initial resistance then the other baths and remained relatively stable throughout the test

### **Temperature Cycle**

Low	Dwell	High	Dwell
Temperatu	Time	Temperatu	Time
re		re	
-65°C	15 (min)	+177°C	15 (min)

### **Resistance Measurements (ohms)**

Bath	Initial	First	Last	Final	Change
		high	high		%
		temp.	temp.		
Α	0.901	1.213	0.882	0.610	28%
В	0.720	0.379	0.427	0.304	11.5%
С	0.165	0.244	0.242	0.161	1%



## **Microsection Analysis**

After thermal shock testing, each coupon was microsectioned and visually inspected. No plating cracks, blistering or delamination was observed in the PTH's.





Bath C





Bath	10 sec.	20 sec.	30 sec.
А	pass	pass	fail
В	pass	fail	fail
С	pass	pass	pass

**Thermal Stress Test Results** 

• Test specimens from each bath were subjected to thermal stress tests consisting of an oven bake out at 121° C for a minimum of six hours and a solder dip at 260° C for 10 sec., 20 sec. and 30 sec.

- Bath A failed at 30 sec.
- Bath B failed at 20 sec.
- Bath C passed all three solder dips.
- All failures were delamination of the plated copper from the dielectric in the thru hole.



# Microsection Analysis of Thermal Shock Coupons



Bath A failed at 30 sec

Bath B failed at 20 sec



Bath C passed all three solder dips





## High Aspect Ratio Through Hole Plating Test

- 91 mil thick polyimide board
- Seven hole sizes of 8, 10, 15, 20, 28, 40 and 50 mils
- Aspect ratios from 1.25:1 up to 11.37

### **Test Coupon**





### **Surface to Hole Thickness Ratio**

Aspect	Bath A	Bath B	Bath C
Ratio			
11.37:1	2.73	1.97	3.03/closed
9.1:1	2.63	1.88	2.98/closed
6.06:1	2.31	1.78	2.78/
4.55:1	1.94	1.68	2.3
3.25:1	1.79	1.62	2.11
2.27:1	1.65	1.58	1.92
1.82:1	1.55	1.43	1.82

• SHTR is determined by dividing the plating thickness on the surface of the hole by the plating thickness in the center of the hole

- The plated copper thickness in the thru holes was 1.25-1.5 mils
- Baths A and B reliably plated hole sizes down to 8 mils
- Bath B had the lowest SHTR

• Bath C had the highest SHTR and did not have the throwing power to deposit the 1.25 mils of copper without closing the 8 and 10 mil holes

# Microsection Analysis of High Aspect Ratio Through Hole Plating Test Coupon



Bath B 8 mil hole

Bath C 8 mil hole



### Bath B 10 mil hole



### Bath C 10 mil hole





## Conclusion

- It was demonstrated that the nanocrystalline deposits produced using PED with grain sizes smaller than 100nm have a lower electrical resistance and a higher hardness then polycrystalline copper deposits produced by DC plating
- The nanocrystalline deposits passed the elevated temperature of the thermal shock test and the increased dwell time of the thermal stress test
- Nanocrystalline plating process with a brightener additive exhibited a higher throwing power when plating high aspect ratio holes
- It can be concluded that nanocrystalline plating processes can be employed in the fabrication of printed wiring boards and they demonstrate superior mechanical and physical properties compared to conventional polycrystalline copper deposits